

# Synthesis and piezoelectric properties of $[\text{Bi}_{1-z}(\text{Na}_{1-x-y-z}\text{K}_x\text{Li}_y)]_{0.5}\text{Ba}_z\text{TiO}_3$ lead-free piezoelectric ceramics

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**Abstract**  $[\text{Bi}_{1-z}(\text{Na}_{1-x-y-z}\text{K}_x\text{Li}_y)]_{0.5}\text{Ba}_z\text{TiO}_3$  lead-free piezoelectric ceramics were fabricated by ordinary ceramic technique and the piezoelectric and ferroelectric properties of the ceramics were studied. The ceramics can be well sintered at 1,100–1,150 °C for 2 h. X-ray diffraction (XRD) analysis shows that  $\text{K}^+$ ,  $\text{Li}^+$  and  $\text{Ba}^{2+}$  diffuse into the  $\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3$  lattices to form a solid solution with a single-phase perovskite structure. The introduction of  $\text{K}^+$ ,  $\text{Li}^+$  and  $\text{Ba}^{2+}$  into  $\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3$  significantly decreases the coercive field  $E_c$  but maintains the large remanent polarization  $P_r$  of the materials. The ceramics provide piezoelectric constant  $d_{33}$  of 205 pC/N, electromechanical coupling factor  $k_p$  of 0.346, remanent polarization  $P_r$  of 31.7–38.5  $\mu\text{C}/\text{cm}^2$ , and coercive field  $E_c$  of 3.18–5.16 kV/mm.

**Keywords**  $[\text{Bi}_{1-z}(\text{Na}_{1-x-y-z}\text{K}_x\text{Li}_y)]_{0.5}\text{Ba}_z\text{TiO}_3$  · Lead-free piezoelectric ceramics · Piezoelectric properties ·  $P$ - $E$  hysteresis loop

## 1 Introduction

$\text{Pb}(\text{Ti},\text{Zr})\text{TiO}_3$ -based piezoelectric ceramics are widely used for various electronic and microelectronic devices. However, the toxicity of lead oxides and the volatilization during sintering process have caused serious environmental problems. To minimize lead pollution, it is necessary to develop environmental-friendly lead-free piezoelectric ceramics.

At present, alkali metal niobate-based, bismuth layer-structured-type,  $\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3$ -based (abbreviated as BNT) and  $\text{BaTiO}_3$ -based ferroelectrics are known as lead-free piezoelectric ceramics. Among these materials, BNT is thought to be a strong candidate of lead-free piezoelectric ceramics because of its relatively large remanent polarization ( $P_r=38 \mu\text{C}/\text{cm}^2$ ) [1]. However, because pure BNT ceramics have a high coercive field ( $E_c=7.3 \text{ kV}/\text{mm}$ ) and large electric conductivity [1], these ceramics are difficult to be completely polarized.

In order to improve the piezoelectric and ferroelectric properties of BNT ceramics, some solid solutions of BNT with  $\text{BaTiO}_3$  [2,3],  $\text{Bi}_{0.5}\text{K}_{0.5}\text{TiO}_3$  [4],  $\text{NaNbO}_3$  [5],  $\text{Ba}(\text{Cu}_{1/2}\text{W}_{1/2})\text{O}_3$  [6],  $\text{BiFeO}_3$  [7] are extensively investigated. But, the piezoelectric constant and the electromechanical coupling factor of these solid solutions are not desirable enough, too. According to the characteristics of BNT ferroelectric, it is believed that a key point to further enhance the piezoelectric properties is to decrease the coercive field efficiently and simultaneously retain the strong ferroelectricity of BNT-based ceramics. Therefore, the research and development of new BNT-based systems with a larger  $P_r$  and lower  $E_c$  may be a promising way to obviously enhance the piezoelectric properties of BNT-based ceramics.

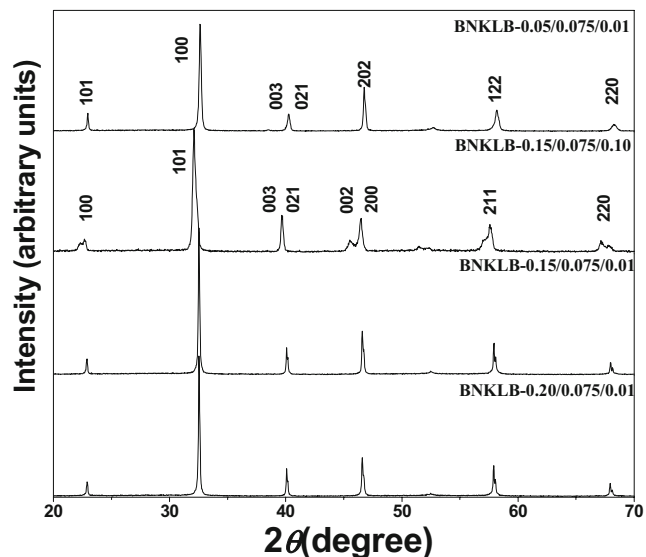
In this paper, new  $[\text{Bi}_{1-z}(\text{Na}_{1-x-y-z}\text{K}_x\text{Li}_y)]_{0.5}\text{Ba}_z\text{TiO}_3$  (abbreviated as BNKLB- $x/y/z$ ) ceramics were prepared by the conventional ceramic processing and their microstructures and piezoelectric properties were studied.

## 2 Experimental

Conventional ceramic fabrication technique was used to prepare BNKLB- $x/y/z$  ceramics. Electronic-grade metal

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**Fig. 1** XRD diffraction patterns of BNKLB- $x/0.075/z$  ceramics in the range  $2\theta$  of 20–70 °

oxides or carbonate powders of  $\text{Bi}_2\text{O}_3$ ,  $\text{Na}_2\text{CO}_3$ ,  $\text{K}_2\text{CO}_3$ ,  $\text{Li}_2\text{CO}_3$ ,  $\text{BaCO}_3$  and  $\text{TiO}_2$  were used as starting raw materials. All the raw materials mixed by ball-milling were calcined at 850 °C for 2 h. After calcination, the ball-milled powders were granulated by adding PVA as a binder and

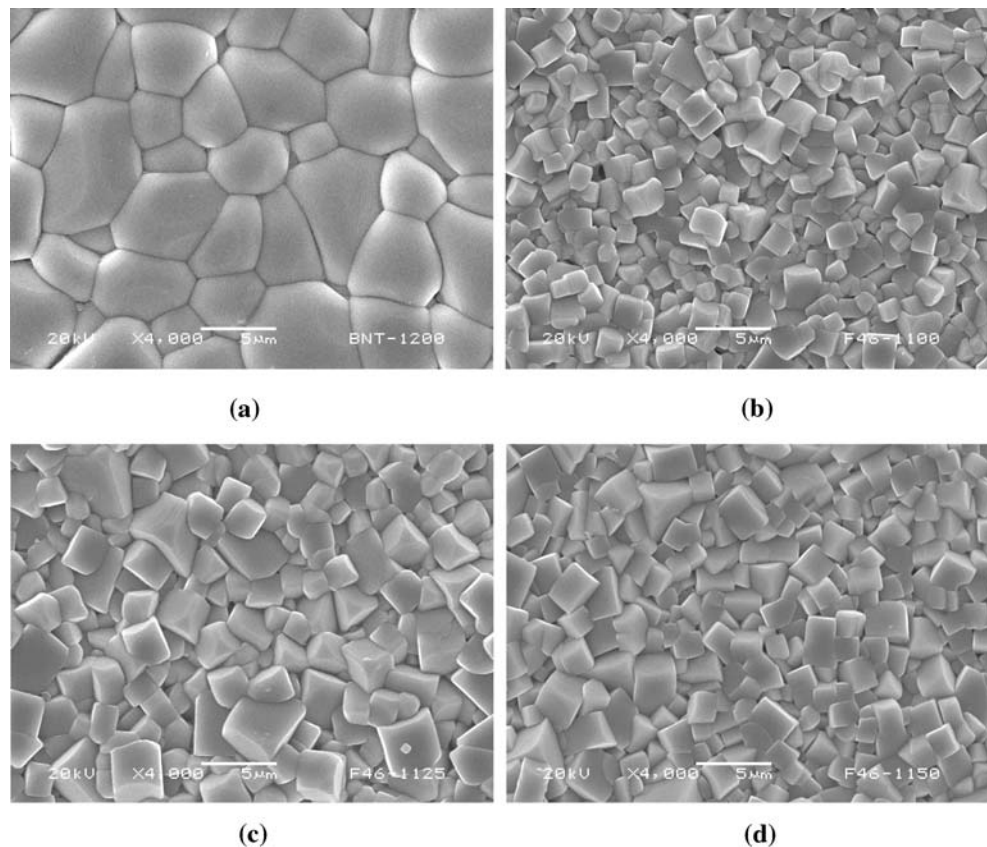
pressed into discs and then sintered at 1,100–1,150 °C for 2 h in air. Silver paste was coated to form electrodes on both sides of sintered samples and the samples were fired at 810 °C for 10 min. The samples were poled in a silicone oil bath with a dc field of 3–4 kV/mm at 80 °C for 20 min.

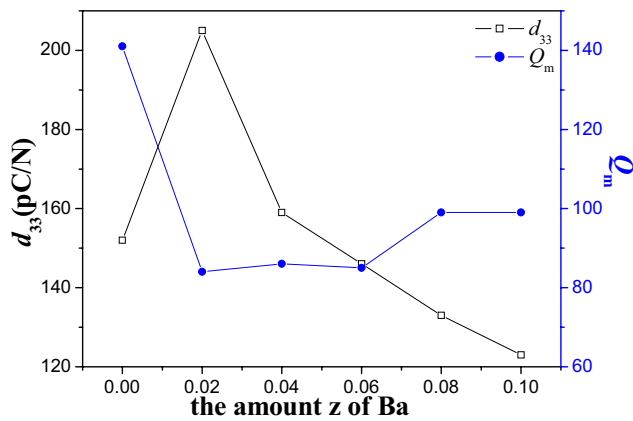
The crystalline structure of the sintered samples was examined by X-ray diffraction technique (DX-1000X, China). The bulk densities of the sintered samples were measured by the Archimedes method. The microstructures of the sintered samples were observed using scanning electron microscope (JSM-5900LV). Planar electromechanical coupling factor  $k_p$  was determined by the resonance-antiresonance method on the basis of IEEE standards using an impedance analyzer (HP4194A). The piezoelectric constant  $d_{33}$  was measured using a piezo- $d_{33}$  meter (ZJ-3A, China). The dielectric properties of the ceramics were measured using a capacitance meter (HP 4287A). The  $P$ - $E$  hysteresis loops were obtained using Radiant Precision Workstation.

### 3 Results and discussion

The XRD diffraction patterns of BNKLB- $x/y/z$  ceramics are shown in Fig. 1. From Fig. 1, the ceramics possess a

**Fig. 2** SEM micrographs of the sintered samples (a) BNKLB-0/0/0 sintered at 1,200 °C for 2 h; (b) BNKLB-0.15/0.075/0.02 sintered at 1,100 °C for 2 h; (c) BNKLB-0.15/0.075/0.02 sintered at 1,125 °C for 2 h; and (d) BNKLB-0.15/0.075/0.02 sintered at 1,150 °C for 2 h

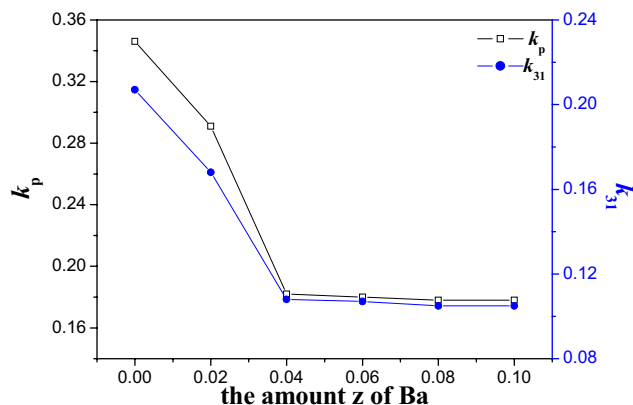




**Fig. 3** Piezoelectric constant  $d_{33}$  and mechanical quality factor  $Q_m$  as a function of  $z$  for BNKLB-0.15/0.075/ $z$  ceramics

single crystalline phase with perovskite structure and no second phase can be found, which reveal that  $K^+$ ,  $Li^+$  and  $Ba^{2+}$  diffuse into  $Bi_{0.5}Na_{0.5}TiO_3$  lattices to form a solid solution. The BNKLN- $x/y/z$  ceramics sintered at 1,100–1,150 °C for 2 h have the bulk densities of 5.62–5.75 g/cm<sup>3</sup> which are 95% higher than the theoretical densities.

The microstructures of the sintered BNKLB- $x/y/z$  ceramics were observed using SEM, and the micrographs of these ceramics are shown in Fig. 2. Pure BNT ceramics sintered at 1,200 °C for 2 h are very dense and the shape of the grains is not regular (Fig. 2(a)). The diffusion of  $K^+$ ,  $Li^+$  and  $Ba^{2+}$  into BNT lattices greatly influences the microstructures of the materials. From Fig. 2 (b)–(d), differently from pure BNT ceramics, the grain growth of BNKLB-0.15/0.075/0.02 ceramics sintered at 1,100–1,150 °C for 2 h is suppressed, and the grains become smaller and grow into neat rectangular shapes. With sintering temperature increasing, the grains of BNKLB-0.15/0.075/0.02 ceramics slightly become larger and more rectangular. At 1,100–1,150 °C for 2 h, the ceramics are very dense and almost no holes can be found.



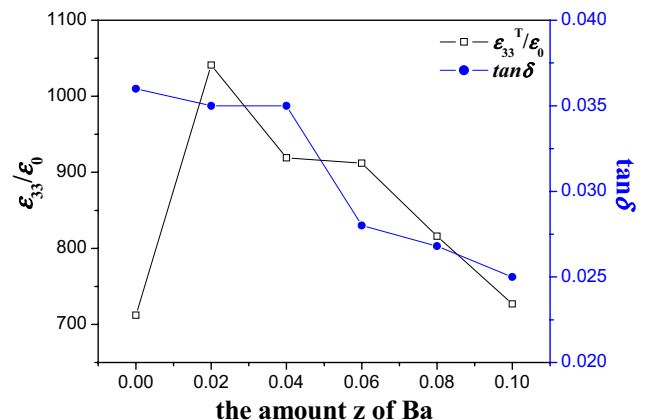
**Fig. 4** Electromechanical coupling factors  $k_p$  and  $k_{31}$  as a function of  $z$  for BNKLB-0.15/0.075/ $z$  ceramics

Figure 3 shows the piezoelectric constant  $d_{33}$  and mechanical quality factor  $Q_m$  as a function of  $z$  for BNKLB-0.15/0.075/ $z$  ceramics. The piezoelectric constant  $d_{33}$  reaches a maximum value of 205pC/N at  $z=0.02$ . The mechanical quality factor  $Q_m$  decreases sharply with  $z$  increasing from 0 to 0.02. When the amount  $z$  of Ba further increases,  $Q_m$  is almost constant.

Electromechanical coupling factors  $k_p$  and  $k_{31}$  as a function of  $z$  for BNKLB-0.15/0.075/ $z$  ceramics are shown in Fig. 4. It can be found that planar electromechanical coupling factor  $k_p$  decreases with  $z$  increasing up to 0.04, and then has no evident change with the further increasing of  $z$ .  $k_p$  reaches the maximum value of 34.6% at  $z=0$ . The variation of the electromechanical coupling factor  $k_{31}$  with  $x$  is similar to that of  $k_p$ .

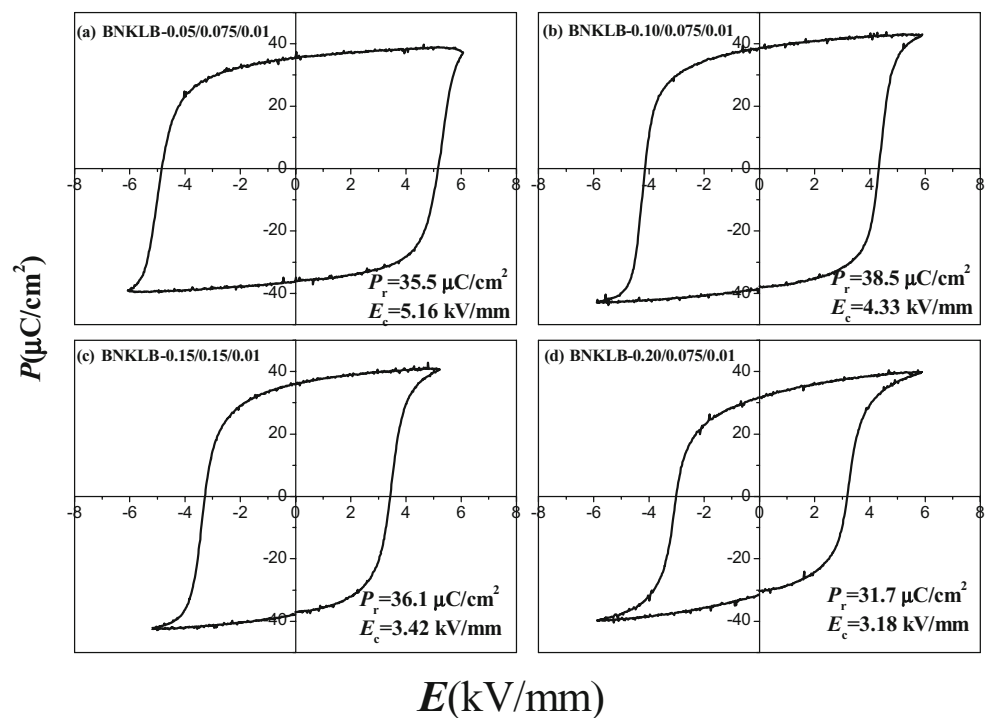
Figure 5 shows dielectric constant  $\epsilon_{33}^T/\epsilon_0$  and dielectric loss  $\tan\delta$  as a function of  $z$  for BNKLB-0.15/0.075/ $z$  ceramics. With the increase of  $x$ , dielectric loss  $\tan\delta$  decreases from 0.036 to 0.025. The dielectric constant  $\epsilon_{33}^T/\epsilon_0$  reaches a maximum value of 1041 at  $z=0.02$ .

The  $P$ - $E$  hysteresis loops of BNKLB- $x/0.075/0.01$  ceramics at room temperature are shown in Fig. 6. The  $P$ - $E$  hysteresis loops are almost rectangular shapes. It is believed that these  $P$ - $E$  hysteresis loops are saturated. With  $x$  increasing from 0.05 to 0.20, the coercive field  $E_c$  decreases from 5.16 to 3.18 kv/mm. The maximum value of remanent polarization  $P_r$  of 38.5  $\mu\text{C}/\text{cm}^2$  appears at  $x=0.20$ . Compared with pure BNT ceramics [8], it is concluded that the partial substitution of  $Na^+$  by  $K^+$  and  $Li^+$  and partial substitution  $(Bi_{0.5}Na_{0.5})^{2+}$  by  $Ba^{2+}$  greatly decrease the coercive field  $E_c$  but simultaneously keep the very high remanent polarization  $P_r$ , which leads to facilitating the poling process of the BNKLB- $x/y/z$  ceramics and an obvious enhancement of piezoelectric properties of the materials. Therefore BNKLB- $x/y/z$  ceramics show better piezoelectric properties.



**Fig. 5** Dielectric constant  $\epsilon_{33}^T/\epsilon_0$  and dielectric loss  $\tan\delta$  as a function of  $z$  for BNKLB-0.15/0.075/ $z$  ceramics

**Fig. 6**  $P$ - $E$  hysteresis loops of BNKLB- $x$ /0.075/0.01 ceramics at room temperature



#### 4 Conclusion

New  $[\text{Bi}_{1-z}(\text{Na}_{1-x-y-z}\text{K}_x\text{Li}_y)]_{0.5}\text{Ba}_z\text{TiO}_3$  ceramics were prepared by the conventional ceramic sintering processing and their microstructures and piezoelectric properties were studied. The ceramics possess a single phase with perovskite structure. The introduction of  $\text{K}^+$ ,  $\text{Li}^+$  and  $\text{Ba}^{2+}$  to the BNT lattice causes an evident change in the microstructure of the BNT ceramics and decreases coercive field greatly but large remanent polarization is simultaneously maintained. The ceramics exhibit piezoelectric constant  $d_{33}$  of 205 pC/N, electromechanical coupling factor  $k_p$  of 0.346, remanent polarization  $P_r$  of 31.7–38.5  $\mu\text{C}/\text{cm}^2$ , and coercive field  $E_c$  of 3.18–5.16 kV/mm.

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